# Optimization of CdS-free non-toxic electron transport layer for Sb<sub>2</sub>S<sub>3</sub>-based solar cell with notable enhanced performance

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#### Abstract

In this investigation, we develop CdS-free non-toxic thin-film solar cell structure with antimony sulfide  $(Sb_2S_3)$  as an absorber material.  $Sb_2S_3$  has found to be a promising candidate for production of renewable energy. Solar cells based on  $Sb_2S_3$  have been attracted worldwide attraction due to their outstanding efficiency and low cost. To serve as an optimistic buffer layer, 3C-SiC (cubic silicon carbide) is used thanks to its suitable bandgap to replace toxic cadmium sulfide (CdS). SCAPS-1D (one-dimensional solar cell capacitance simulator) software has been employed to numerically investigate the performance of  $Sb_2S_3$ -based n-ZnO/n-3C-SiC/p-Sb\_2S\_3 heterostructure solar cells. The influence of absorber/buffer layer thickness, acceptor/ donor densities, and defect density on device working have been investigated. Consequently, the role of defects in p-Sb\_2S\_3 along with the significance of n-3C-SiC/p-Sb\_2S\_3 interface defects has been studied to provide recommendations for achieving high efficiency. The proposed structure provides the enhanced efficiency of 17% under 1.5 G illumination spectrum. The parameters regarding solar cell performance such as  $V_{oc}$ ,  $J_{sc}$ , FF, QE and  $\eta$  have been studied graphically. This novel structure may have considerable influence on progress of improved photovoltaic devices in future.

**Keywords**  $3C-SiC \cdot Sb_2S_3 \cdot Buffer layer \cdot Efficiency \cdot Absorber layer \cdot SCAPS-1D$ 

### 1 Introduction

Photovoltaic devices have attracted a huge attention worldwide for successive energy harvesting due to several benefits including everlasting power at least working cost, long lifetime, very less pollution, lower maintenance and easy to use [5]. Thin-film solar cells are the most prominent candidate in photovoltaics due to their low-cost production, lower maintenance, high efficiency, user friendly and most importantly low pollution [6–12]. In the production of commercial

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TFSCs, many materials are used as absorbing layers like silicon [13–17], Cu(In,Ga)Se<sub>2</sub> [18, 19], Cu<sub>2</sub>ZnSn(S,Se)<sub>4</sub> [20–22] CdTe [23–26]. However, the problem remains in the manufacturing of highly efficient TFSCs using these absorbing layers due to high cost, low abundancy, and high toxicity. Several investigations have been made to overcome these challenges by finding new absorbing materials in the application of TFSCs that have properties such as highly abundance, cost-efficiency, environment-friendly, and high stability [27–31]. Selenium (Se) has recently gained interest of global research groups due to its wide bandgap (1.8 eV) and intrinsic environmental stability [32]. Though, the melt processing limits the use of Se to some extents due to its deprived wettability which melt Se on top of some materials such as TiO<sub>2</sub>. Antimony Sulfide  $(Sb_2S_3)$  is a newly found optimal absorbing material in the application of TFSCs due to good career mobility [33-37], high abundancy, ideal absorption coefficient  $(10^5 \text{ cm}^{-1})$ , propitious range of band gap (1-1.2eV), cost-efficient and less toxic [33, 35, 38–41]. In several theoretical [42–47] and experimental [33, 37, 38, 48–55], the auspicious performance of p-Sb<sub>2</sub>S<sub>3</sub> as absorbing material has been reported. The performance efficiency of 7.23% was achieved by employing liquid medium annealing



[1]. With device scheme of ITO/TiO<sub>2</sub>/Zn(O,S)/Sb<sub>2</sub>S<sub>2</sub>/Carbon/Ag, 3.7% efficiency has been achieved by using in situ hydrothermal deposition [2]. 8% efficiency was obtained by using unique chemical bath deposition technique which improves the deposition rate, crystallinity and film morphology as well [3]. Vertically oriented thin-films obtained on various buffer layers and 4.37% efficiency have been achieved with structure (ITO)/CdS/Sb<sub>2</sub>S<sub>3</sub>/Au [4]. The 2.26% maximum efficiency was achieved by using TiO<sub>2</sub> as buffer layer in FTO/TiO<sub>2</sub>/Sb<sub>2</sub>Se<sub>3</sub>/Au-based solar cells [38]. On experimental basis, heterojunction solar cell with Sb<sub>2</sub>Se<sub>3</sub> 3.21% efficiency was achieved [48]. The PCE (%) of 2.1% was obtained by using glass/-FTO/TiO<sub>2</sub>/Sb<sub>2</sub>Se<sub>3</sub>/CuSCN/ Au-based solar cells, where HTL was CuSCN [50]. The maximum conversion efficiency of 5.6% was achieved by applying thermal evaporation method in glass/-FTO/CdS/ Sb<sub>2</sub>Se<sub>3</sub>/Au-based solar cells [33]. The PCE (%) of 7.6% was obtained with ITO/CdS/Sb<sub>2</sub>Se<sub>3</sub>/Au solar cells that were designed by vapor transport deposition technique [54]. Germanium mono-selenide (GeSe), another promising material, has been used to obtain the efficiency of 3.1% with device configuration of Glass/ITO/CdS/GeSe/Au. A substrate special arrangement adopted that allows CdS pre-deposition on top of polycrystalline GeSe layer at room temperature and avoid the diffusion of Cd [56]. A suitable buffer layer or window can be chosen on the basis of high optical transmittance and electric conductivity which enhance the efficiency but also protect it from damages [57]. In this respect, cubic silicon carbide (3C-SiC) is one of the most prominent candidates as a buffer material or window in TFSCs application due to its bandgap of 2.36 V [58]. The great electronic properties of 3C-SiC have been reported in comparison with other buffer materials like SiC. The distinctive property of 3C-SiC is that it has  $10^{16}$  cm<sup>-3</sup> of intrinsic concentration at 300 K and can be n- doped with Ga, Al, and B dopants. [59–61]. High efficiency of 25.51% has been reported in a recent simulation study in CIGS TFSCs which was modeled with a buffer layer or window with 3C-SiC [62]. Therefore, the performance of heterojunction solar cell-based 3C-SiC/  $Sb_2S_3$  is investigated by the theoretical approach.

In this study, the main objective is to study the feasibility of 3C-SiC/Sb2S3 heterojunction photovoltaic cell. The simulation tool was used for the vast and complex optimized design of the  $Sb_2S_3$ -based photovoltaic cell. The Scaps-1D has been utilized to enhance chronologically the configuration of the photovoltaic cell by exploring defect density, outcome of doping concentration, and width of buffer layer or window 3C-SiC and absorber  $Sb_2S_3$ . To prevent any degradation in analyzing the solar cell operation, changes in operating temperature have been made in the range.

This paper consists of methodology, results, and conclusion. In the methodology section, we introduced the parameter values, model, and formulation of heterostructure solar cells by using simulation in SCAPS-1D. In the result section, we plotted the parameters of proposed structures such as Fill factor (FF), Short-circuit current density  $(J_{sc})$ , open-circuit voltage  $(V_{oc})$ , and efficiency ( $\eta$ ). The last section is based on a conclusion and summary.

### 2 Methodology

One-dimensional solar cell capacitance simulator (SCAPS-1D) [63] has been utilized to develop and examine the  $Sb_2S_3$ -based photovoltaic device. It solves continuity and Poisson equations for electrostatic potential and free carriers (holes/electrons) for heterojunction devices. These equations are given below,

$$\frac{\partial}{\partial x} \left( \varepsilon_0 \varepsilon \frac{\partial \Psi}{\partial x} \right) = -q \left( p - n + N_D^+ - N_A^- + \frac{\rho_{def}}{q} \right) \tag{1}$$

$$-\frac{\partial J_n}{\partial x} - U_n + G = \frac{\partial n}{\partial t}$$
(2)

$$-\frac{\partial p}{\partial x} - U_p + G = \frac{\partial p}{\partial t}$$
(3)

Here,  $\Psi$  is the electric potential,  $\varepsilon$  is permittivity, charge represented by q,  $N_{\rm a}$  and  $N_{\rm d}$  are the acceptor and donor densities, p and n are holes and electron densities, G is the generation rate, and  $J_p$  and  $J_n$  are the current densities for holes and electrons, respectively. The performance of device limitations such as recombination profile, J-V (current density-voltage) characteristics, energy band diagram and quantum efficiency can be defined by solutions of equations. The numerical simulation is carried out under 100 mW cm<sup>-2</sup> illumination, with 300 K temperature and air mass AM 1.5 G to analyze the working of device. In the present work, 3C-SiC buffer layer has been introduced as the replacement of CdS layer. The comparison has been made to analyze the design of optimized heterostructure solar cell. The representation of Sb<sub>2</sub>S<sub>3</sub>-based heterojunctions of Al/ZnO/3C-SiC/Sb<sub>2</sub>S<sub>3</sub>/ Mo is shown in Fig. 1b.

The structure contains window layer of ZnO with energy bandgap of 3.30 eV, the buffer layer of n-type 3C-SiC with bandgap of 2.62 eV [58] and p-Sb<sub>2</sub>S<sub>3</sub> as absorber with bandgap of 1.62 eV [64]. The transfer of hole and electron beyond the heterostructure can be significantly determined by band structure analysis. SCAPS-1D offers the energy band configuration for the suggested heterostructure of device. The related parameters to material characteristics employed in this work were extracted from experimental and theoretical studies reported earlier [58, 64, 65] and also described in Table 1. The factors such as layer thickness, defect density and doping concentration of buffer and absorber layer and 3C-SiC buffer layer



Table 1 Parameters affecting the device performance and simulation values

| Factors (unit)                      | ZnO                  | 3C-SiC               | $Sb_2S_3$              |
|-------------------------------------|----------------------|----------------------|------------------------|
| Thickness w(µm)                     | 0.05                 | 0.05                 | 2.0                    |
| $E_g(eV)$                           | 3.30                 | 2.36                 | 1.62                   |
| χ(eV)                               | 4.6                  | 3.38                 | 3.7                    |
| ε <sub>r</sub>                      | 8.9                  | 9.72                 | 7.08                   |
| $N_c ({\rm cm}^{-3})$               | $2.2 \times 10^{18}$ | $1.5 \times 10^{19}$ | $2 \times 10^{19}$     |
| $N_{v} ({\rm cm}^{-3})$             | $1.8 \times 10^{19}$ | $1.2 \times 10^{19}$ | $1 \times 10^{19}$     |
| Thermal velocity (electron cm/s)    | $1 \times 10^{7}$    | $2 \times 10^{7}$    | $1 \times 10^{7}$      |
| Thermal velocity (hole cm/s)        | $1 \times 10^{7}$    | $1.5 \times 10^{7}$  | $1 \times 10^{7}$      |
| $\mu_n (\mathrm{cm}^2/\mathrm{Vs})$ | 100                  | 400                  | 9.8                    |
| $\mu_p(\text{cm}^2/\text{Vs})$      | 25                   | 50                   | 10                     |
| $N_{\rm D} (cm^{-3})$               | 10 <sup>19</sup>     | 1016                 | 0                      |
| $N_A (cm^{-3})$                     | 0                    | 0                    | $5.752 \times 10^{15}$ |
|                                     |                      |                      |                        |

and absorber-buffer interface have been changed involving the device structure. The variables defining performance of device such as J-V characteristics, band profile and quantum efficiency (QE) are obtained from the findings and improved. The influence of Sb<sub>2</sub>S<sub>3</sub> absorber and 3C-SiC layer thickness was analyzed. Furthermore, the influence of operating temperature on performance of device has been also examined.

Energy band diagram for Sb<sub>2</sub>S<sub>3</sub>-based solar cell along with 3C-SiC and CdS buffer layers is demonstrated in Fig. 2b and a, respectively. In Fig. 2a, one can notice a "spike" at Sb<sub>2</sub>S<sub>3</sub>/CdS junction which is a positive conduction band offset ( $\Delta Ec$ ). The lower portion of conduction band gets closer to the Fn (quasi-fermi level) for electrons due to the spike as shown in Fig. 2a. It needs a larger bending in valence band which leads in induction of wider barrier for the holes reaching the interface. In addition, at CdS/ZnO interface the negative conduction band offset increases the chance of recombination which directly affects the device performance.



Fig. 2 Bandgap schema of Sb<sub>2</sub>S<sub>3</sub>-based heterojunction solar cell with buffer layer (3C-SiC)

### 3 Results

The key motive of this study is to examine the effect of several aspects of absorber and buffer width on working of  $Sb_2S_3$  based photovoltaic cells. Application of the improved data will assist us to resolve a set of constraints for real time model of solar cell with an optimum conversion efficiency. The simulation and design were assembled for optimization of defect density, thickness, and doping concentrations of *n*-3C-SiC buffer and *p*-Sb<sub>2</sub>S<sub>3</sub> absorber layer. The conduction band offset at the 3C-SiC/Sb<sub>2</sub>S<sub>3</sub> junction is due to difference between the electron affinities of 3C-SiC and Sb<sub>2</sub>S<sub>3</sub> materials. In this way, photoelectrons move toward the 3C-SiC layer from *p*-Sb2S3 before recombination with the holes.

A comparison of simulation results has been made for CdS and 3C-SiC buffered solar cell in Fig. 3. It represents the analysis of J–V (current density and voltage) characteristics and Spectral response for proposed structure. The thicknesses of Sb<sub>2</sub>S<sub>3</sub>, 3C-SiC and ZnO layers are 2  $\mu$ m, 0.05  $\mu$ m and 0.05  $\mu$ m, respectively. One can notice that the



Fig.3 Represents the  $a\ J{-}V$  analysis,  $b\ Quantum$  efficiency of CdS and 3C-SiC buffered solar cell

voltage and current density for proposed heterostructure with 3C-SiC buffer layer is good than the device with CdS layer. The  $V_{\rm oc}$  with CdS layer is determined to be 1.08 V with  $J_{\rm sc}$ of 22.64 mA/cm<sup>2</sup>, FF of 58% and efficiency of 14%, whereas the enhanced  $V_{\rm oc}$  of 1.4 V,  $J_{\rm sc}$  of 23.52 mA/cm<sup>2</sup>, FF 47% and efficiency of 17.18% has been obtained for the PV cell structure with 3C-SiC layer. The spectral response (quantum efficiency) of both device structures is illustrated in Fig. 2b. The wavelength is changing from 300 to 800 nm to calculate spectral response. It is clear from results that change of buffer layer does not affect the spectral response of solar cell significantly. However, it can be improved by introducing a hole transport layer (HTL) or back surface field (BSF) layer at rear of device [66] which would be incorporated in future investigations. The spectral response attained at the range of short wavelength in this work, which is in good agreement with previous studies reported earlier [66-68]. By introducing 3C-SiC buffer layer, a significant improvement in the solar cell performance parameters has been realized. Therefore, all the simulations by changing parameters are carried out for ZnO/3C-SiC/Sb<sub>2</sub>S<sub>3</sub>/Mo heterostructure.

#### 3.1 Optimization of buffer layer (3C-SiC) thickness

The thickness of 3C-SiC layer changes to analyze the functioning of Sb<sub>2</sub>S<sub>3</sub> solar cell. The width of buffer layer varies from 10 to 100 nm with 10 nm each step size. The buffer layer should be narrower to decrease the impact of series resistance [69]. The influence of 3C-SiC layer thickness on  $\eta$ , FF,  $V_{oc}$  and  $J_{sc}$  has been analyzed, as displayed in Fig. 4a, b. The highest efficiency of 17% has been achieved for layer thickness of 0.05  $\mu$ m, while  $J_{sc}$  firstly show a minor increase and then, very slight decrease from 23.5 to 23.4 mA/cm<sup>-2</sup>. The  $V_{\rm oc}$  increases with a large step from 1.1 to 1.5 V at start; with further rise in buffer layer width, it shows very minor change in the entire range. The nearly constant  $J_{sc}$  value can be explained as the thickness of 3C-SiC layer rises which give growth in absorption of photons at more distance of  $Sb_2S_3/3C-SiC$  junction with shorter wavelength. It reduces intensity of photons which are strong enough to penetrate to  $Sb_2S_3$  layer [70–72]. Figure 4b illustrates that FF noticeably decreases from 55 to 47% as thickness of increases from 10 to 20 nm of buffer layer. Further increment in thickness of buffer layer does not affect the FF, so the optimized value for layer thickness is considered 0.05  $\mu$ m. FF and  $J_{sc}$  have an inverse relationship; both parameters trivially vary in opposite trends as thickness increases. The rise in the thickness of 3C-SiC layer has a negative impact on efficiency, indicating the increase in absorption of photons in the buffer layer. As a result, fewer number of photons approached to the Sb<sub>2</sub>S<sub>3</sub> layer which led to decrease the production of electron-hole pair. The overall performance of the Sb<sub>2</sub>S<sub>3</sub>-based solar cell improves with the addition of a 3C-SiC layer as a buffer



**Fig. 4** Optimization of buffer layer (3C-SiO) thickness in terms of **a**  $V_{oc}$  and  $J_{sc}$  **b** Efficiency and FF

layer instead of CdS, but all other parameters showed insignificant change with change in the thickness of buffer layer.

### 3.2 Optimization of doping concentration of buffer layer (3C-SiC)

To investigate the influence of donor doping concentration  $(N_D)$  in 3C-SiC layer on solar cell performance,  $N_D$  varies from  $10^{12}$  to  $10^{18}$  cm<sup>-3</sup> as demonstrated in Fig. 5. The value of  $V_{\rm oc}$  is almost constant in the range of  $10^{12}$ – $10^{16}$  $cm^{-3}$ ; then, it increases with the further increase in donor concentration. As a result, increase in non-recombination losses, causing the saturation  $J_{sc}$  to rise. Therefore, the value of  $V_{\rm oc}$  significantly increases as the donor concentration varies [73]. Figure 5a also shows how  $J_{sc}$  responds to variations in the doping concentration of the 3C-SiC layer. With the increase in donor concentration, the  $J_{sc}$  exhibits the similar trend as of  $V_{\rm oc}$ , it remains unchanged for the range of N<sub>D</sub> from  $10^{12}$  to  $10^{16}$  cm<sup>-3</sup> and then, slightly increases due to reduction in carrier diffusion length [74]. Figure 5b shows the variation in FF as donor concentration changes. FF exhibits notable increment from 53.3 to 57.4% from  $10^{16}$  to  $10^{17}$  cm<sup>-3</sup> and rapidly decreases for higher values



**Fig. 5** Optimization of donor density ( $N_D$ ) of buffer layer (3C-SiO) in terms of **a**  $V_{oc}$  and  $J_{sc}$  **b** Efficiency and FF

of donor concentration till  $10^{18}$  cm<sup>-3</sup>. This behavior is in view of the fact that large carrier concentration increases recombination losses at junction [73]. Figure 5b illustrated the efficiency against the doping concentrations of buffer layer. The efficiency increases monotonically as the doping concentration increases from till  $10^{16}$  to  $10^{18}$  cm<sup>-3</sup>. Hence, the optimal value of doping concentration for buffer layer is  $1.0 \times 10^{18}$  cm<sup>-3</sup>.

# 3.3 Optimization of thickness of absorber layer (p-Sb<sub>2</sub>S<sub>3</sub>)

The thickness of absorber layer (Sb<sub>2</sub>S<sub>3</sub>) plays a significant aspect to achieve the maximum PCE (%) of 3C-SiC/Sb<sub>2</sub>S<sub>3</sub> heterojunction solar cell. The influence of changing absorber layer thickness on  $J_{sc}$ ,  $V_{oc}$ ,  $\eta$  and FF is analyzed and demonstrated in Fig. 6. The  $J_{sc}$  and  $V_{oc}$  increase as the absorber layer thickness increases from 1.0 to 2.0 µm. The  $V_{oc}$  and  $J_{sc}$  show a rapid increment till 2.0 µm and then, slightly increases for higher values, which shows a non-significant behavior as demonstrated in Fig. 6a. The efficiency shows almost the same behavior with the increase in absorber layer thickness from 1.0 to 2.0 µm demonstrated in Fig. 6b. One can notice



**Fig. 6** Optimization of absorber layer (Sb<sub>2</sub>S<sub>3</sub>) thickness in terms of **a**  $V_{oc}$  and  $J_{sc}$  **b** Efficiency and FF

the decreasing trend of FF from 53.61 to 47.43% for initial increase in layer thickness from 1.0 to 2.0 µm, and non-significant variation for further higher value of layer width. However, the efficiency rapidly rises from 16 to 17% with change in thickness from 1.0 to 2.0 µm of absorber layer. For further increase in p-Sb<sub>2</sub>S<sub>3</sub> layer up to 2.0  $\mu$ m does not affect the efficiency significantly. This enhancement in the device performance is due to more absorption of photon and hole-electron production in the  $p-Sb_2S_3$  layer. When the thickness exceeds the optimal value of absorber layer, the path length moved by photo-generated carriers becomes very large, resulting in more recombination. The optimal value of thickness is 2.0 µm for absorber layer for efficient and low-cost heterojunction solar cell. The results show good agreement with earlier reported results [75–78]. The carrier generation becoming less effective than carrier recombination for thick absorber which is responsible for  $J_{sc}$  saturation [64].

# 3.4 Optimization of doping concentration of absorber layer (Sb<sub>2</sub>S<sub>3</sub>)

The impact of variation in acceptor doping concentration (N<sub>A</sub>) of absorber layer ( $p-Sb_2S_3$ ) on 3C-SiC/Sb<sub>2</sub>S<sub>3</sub>



**Fig. 7** Optimization of acceptor density  $(N_A)$  for absorber layer  $(Sb_2S_3)$  in terms of **a**  $V_{oc}$  and  $J_{sc}$  **b** Efficiency and FF

heterostructure device is demonstrated in Fig. 7. Higher values more than 10<sup>15</sup> and lower values less than 10<sup>14</sup> distinctively affect the optoelectronic properties of absorber layer. Highest efficiency for absorber layer of about 17.3% is recorded when doping concentration is  $10^{15}$  cm<sup>-3</sup>. Efficiency increases linearly when  $N_A$  rises from  $10^{13}$  cm<sup>-3</sup> because there is an increase in carrier concentration generation as a result of an increase in doping concentration. [79]. The value of current density ( $J_{sc} = 24.38 \text{ mA/cm}^2$ ) at  $10^{15} \text{ cm}^{-3}$ and continuous drop to 22.12 mA/cm<sup>2</sup> at  $10^{18}$  cm<sup>-3</sup> has been observed. With an increase in acceptor density, holes become more prevalent in the absorber layer, generating the hole trap center within the absorber layer. This enhances photo-generated electron recombination due to the attraction between the hole and electron, as a result the current density decreases as acceptor concentration increases.

Rashed et al. [76] for Sb<sub>2</sub>S<sub>3</sub> heterojunction solar cell and Wanda et al. [71] for CZTS-based solar cell previously reported a similar behavior with modification of N<sub>A</sub> in absorber layer. FF and efficiency increase as the acceptor concentration in absorber layer increases as illustrated in Fig. 7a. FF shows a similar behavior of reduction as  $J_{sc}$  for N<sub>A</sub> more than 10<sup>15</sup> cm<sup>-3</sup>. The linear increase in solar cell efficiency is caused by the rising trends in  $V_{oc}$  along with the increase in acceptor concentration of absorber layer.

### 3.5 Optimization of temperature on ZnO/3C-SiC/ Sb<sub>2</sub>S<sub>3</sub> solar cell

The operating temperature is responsible for controlling photovoltaic cell performance. When the operating temperature is high, the carrier concentration rises that results in increasing the rate of internal carrier recombination. Therefore, it is typically observed that device performance decreases as temperature rises [80–82]. As indicated in the literature, the temperature has a significant impact on the stability of solar cell, with the intent that open-circuit voltage varies with temperature change [83]. In this study, the operating temperature has been adjusted between 300 and 400 K to examine device performance as illustrated in Fig. 8. The value of  $V_{\rm oc}$  drops as the temperature increases due to rise in recombination as shown in Fig. 8a [81, 84]. With the increase in temperature, value of  $V_{\rm oc}$  linearly decreases from 1.542 to 0.961 V and  $J_{sc}$  does not show a significant variation a minor increase from 23.5 to 23.7 mA/cm<sup>2</sup> has been noticed. High temperatures reduce the bandgap of material and generate significant band-to-band absorption coefficients across the spectrum [80, 81] increasing leakage current and resulting in performance degradation of device. The impact of variation of temperature on FF has been analyzed as illustrated in Fig. 8c. The increment in cell temperature has a unique influence on cell parameters like FF,  $V_{oc}$  and  $\eta$ . FF increases as the operating temperature rises due to improving collisions and carrier recombination [84]. Figure 8d depicts the degradation in  $\eta$  with an increment in temperature, which is primarily due to a decrease in bandgap [83]. The parameters like  $\eta$ , FF,  $V_{oc}$  and  $J_{sc}$  validate that the

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proposed ZnO/3C–SiC/Sb<sub>2</sub>S<sub>3</sub> solar cell can work efficiently at 400 K high operating temperature.

## 3.6 Optimization of total defect density of absorber layer (Sb<sub>2</sub>S<sub>3</sub>)

The solar cell performance has significantly influenced by defect density (Nt). The current produced by incident photons is significantly produced due to p-Sb<sub>2</sub>S<sub>3</sub> (absorber layer). Therefore, when defect density rises, carries recombination also rises, which has an impact on the device efficiency. To investigate the variation in device performance, we vary the defect density from  $10^{12}$  to  $10^{16}$  cm<sup>-3</sup> (see Fig. 9). All parameters  $(J_{sc}, V_{oc}, and \eta)$  begin to decrease for defect density values greater than  $10^{15}$  cm<sup>-3</sup> as demonstrated in Fig. 9. However, one can notice that FF starts increasing particularly at  $10^{14}$  cm<sup>-3</sup>, which is due to increase in energy barrier height [85]. The device performance is particularly sensitive to defect density because it generates multiple recombination centers, which improves carrier recombination and consequently reduces carrier lifetime. These recombination centers reduce the production of e-h by preventing the carriers from accessing the junctions. Hence, the overall performance of purposed solar cell decreases. The results are consistent with previous literature [78, 86].

### 4 Conclusion

This study has generated a graphical depiction through numerical analysis. The 3C-SiC has been shown to be a non-toxic prospective material for buffer layer in  $Sb_2S_3$ -based heterojunction solar cells. The SCAPS-1D simulator has been used to simulate the ZnO/3C-SiC/Sb<sub>2</sub>S<sub>3</sub>-based solar



**Fig. 8** Optimization of operating temperature in terms of **a**  $V_{\rm oc}$ , **b**  $J_{\rm sc}$  **c** FF, **d** Efficiency



cell. Device performance has been investigated by analyzing different parameters like thickness of layers, doping concentrations, operating temperature, and defect density. The impact of buffer layer (3C-SiC) and absorber layer  $(Sb_2S_3)$  thickness on FF,  $V_{0c}\eta$ , and  $J_{sc}$  of solar cell has been analyzed. To provide guidance for achieving optimum conversion efficiency, the defects come across in 3C-SiC and  $Sb_2S_3$  layers as well as the role of  $3C-SiC/Sb_2S_3$  interface defects density have been thoroughly examined. The results demonstrate that the efficiency of solar cell with thin buffer layer (3C-SiC) has been improved. The thickness of 3C-SiC layer from 20 to 100 nm was varied to check its viability for optimum conversion efficiency. Maximum 17% conversion efficiency has been obtained with proposed solar cell which contain of ZnO/3C-SiC/Sb<sub>2</sub>S<sub>3</sub> layers under 1.5 G illumination spectrum. The proposed structure could make a significant contribution in the field of photovoltaic devices in the future.

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**Data availability** Enquiries about data availability should be directed to the authors.

#### Declarations

**Conflict of interest** The authors have not disclosed any conflict of interest.

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